

Data collection: *CAD-4 Software* (Enraf–Nonius, 1989). The scan range was  $(0.8 + 0.35 \tan\theta)^\circ$  with a rate of  $0.8\text{--}5.5^\circ \text{ min}^{-1}$ . Background counts were recorded for 25% of the range below and above each peak with a background/scan time ratio of 1/2. Cell refinement: *CAD-4 Software*. Data reduction, structure solution, structure refinement, molecular graphics and software used to prepare material for publication: *MolEN* (Fair, 1990).

Lists of structure factors, anisotropic displacement parameters and complete geometry have been deposited with the IUCr (Reference: BR1068). Copies may be obtained through The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

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*Acta Cryst.* (1994). **C50**, 1377–1379

## Monoclinic $\text{Y}_2\text{Al}_3\text{Si}_2$ with a New Structure Type

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(Received 5 October 1993; accepted 16 March 1994)

### Abstract

The structure of the title compound, trialuminium diyttrium disilicide, can be derived from the  $\text{Au}_2\text{Mn}_5$  structure. The coordination polyhedron of Y has composition  $[\text{Si}_5\text{Al}_7\text{Y}_5]$ , those of the two Al sites have compositions  $[\text{Si}_2\text{Al}_5\text{Y}_5]$  and  $[\text{Si}_2\text{Al}_6\text{Y}_4]$ , and that of Si has composition  $[\text{SiAl}_3\text{Y}_5]$ .

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### Comment

The ternary system Y–Al–Si at 773 K contains six compounds, four with known structures and compositions, *i.e.*  $\text{YAl}_2\text{Si}_2$  [ $\text{La}_2\text{O}_2\text{S}$  *anti*-type structure (Murav'eva, Zarechnyuk & Gladyshevskii, 1971)],  $\text{YAl}_{2.8}\text{Si}_{0.2}$  [ $\text{Mg}_3\text{Cd}$  *anti*-type structure (Murav'eva, Zarechnyuk & Gladyshevskii, 1971)],  $\text{YAl}_{1.4}\text{Si}_{0.6}$  [own structure type (Yanson, 1975)] and  $\text{Y}_6\text{Al}_3\text{Si}$  [ $\text{Tb}_6\text{Al}_3\text{Si}$  type structure (Dubenko, Evdokimov & Titov, 1985)], and two with unknown structures and approximate compositions, *i.e.*  $\text{YAl}_2\text{Si}$  (Yanson, 1975) and  $\text{YAl}_{1.1}\text{Si}_{0.9}$  (Murav'eva, Zarechnyuk & Gladyshevskii, 1971). Here we report the structure of ' $\text{YAl}_2\text{Si}$ ', which has the refined composition  $\text{Y}_2\text{Al}_3\text{Si}_2$ .

The structure can be considered to be a strongly distorted substitution variant of the  $\text{Au}_2\text{Mn}_5$  type structure (Humble, 1964), which itself is an order substitution variant of cubic close packing, but should be considered as a new type. It contains the following coordination polyhedra (see Fig. 1): one Y-centred 17-fold polyhedron of composition  $[\text{Si}_5\text{Al}_7\text{Y}_5]$ , two Al-centred 12-fold polyhedra of compositions  $[\text{Si}_2\text{Al}_5\text{Y}_5]$  for Al(1) and  $[\text{Si}_2\text{Al}_6\text{Y}_4]$  for Al(2), and one Si-centred 9-fold polyhedron of composition  $[\text{SiAl}_3\text{Y}_5]$ . The polyhedron around Y can be described as a pentagonal prism formed by two Y, four Al and four Si atoms, with two Y atoms

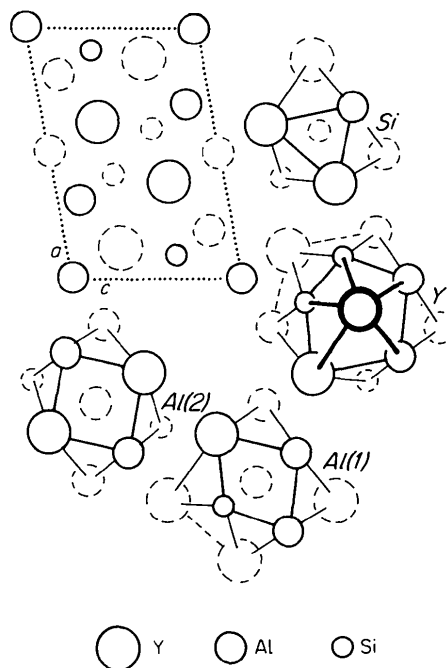


Fig. 1. Structural projection of monoclinic  $\text{Y}_2\text{Al}_3\text{Si}_2$  along  $[010]$ , showing coordination polyhedra. Full lines and circles are at height  $\frac{1}{2}$ , dashed lines and circles are at height 0.

capping the pentagonal faces and one Y, three Al and one Si capping the rectangular faces. The two polyhedra around Al are distorted cuboctahedra, whereas the polyhedron around Si is a distorted trigonal prism formed by four Y and two Al atoms which is capped by one Y, one Al and one Si. Similar polyhedra are found in ternary aluminides containing rare earth and transition elements. In these compounds the transition element often plays the role of a non-metal such as B or Si. For example, in Pr<sub>2</sub>Ni<sub>2</sub>Al (W<sub>2</sub>CoB<sub>2</sub> *anti*-type structure) Ni-centred trigonal [Pr<sub>4</sub>Al<sub>2</sub>] prisms are interconnected *via* common Pr faces and Al edges (Rykhali', Zarechnyuk & Kuten', 1978). In the structures of Al-rich compounds, *e.g.* YNiAl<sub>3</sub> (Gladyshevskii & Parthé, 1992) and YNiAl<sub>4</sub> (Rykhali', Zarechnyuk & Yarmolyuk, 1972), Ni-centred trigonal prisms have the composition [Y<sub>2</sub>Al<sub>4</sub>] and are isolated. In the

present structure the Si-centred trigonal prisms are linked in pairs *via* common Y faces, giving rise to relatively short Si—Si distances.

The bond distances in Y<sub>2</sub>Al<sub>3</sub>Si<sub>2</sub> are in the usual range for metal-rich aluminides and silicides. Noteworthy are the short contacts within the Si pairs (Si—Si = 2.38 Å) and within the Al chains [Al(1)—Al(1) = 2.78 Å] running along [010]. An alternative way to describe the structure of Y<sub>2</sub>Al<sub>3</sub>Si<sub>2</sub> is as slabs (as shown in Fig. 2*a*) mutually shifted by *b*/2. The structural analogy with U<sub>3</sub>Si<sub>2</sub> (Zachariasen, 1948) becomes apparent (Fig. 2*b*) from the way in which these slabs are intergrown.

### Experimental

Crystals were obtained *via* arc melting and annealing at 773 K for three weeks.

#### Crystal data

Y<sub>2</sub>Al<sub>3</sub>Si<sub>2</sub>  
*M<sub>r</sub>* = 314.93  
 Monoclinic  
*C2/m*  
*a* = 10.220 (3) Å  
*b* = 4.0354 (9) Å  
*c* = 6.617 (2) Å  
 $\beta$  = 101.36 (3)°  
*V* = 267.54 (7) Å<sup>3</sup>  
*Z* = 2  
*D<sub>x</sub>* = 3.909 Mg m<sup>-3</sup>

Mo *K*α radiation  
 $\lambda$  = 0.71073 Å  
 Cell parameters from 25 reflections  
 $\theta$  = 7.5–17°  
 $\mu$  = 22.357 mm<sup>-1</sup>  
*T* = 293 K  
 Needle  
 0.080 × 0.020 × 0.012 mm  
 Silverish

#### Data collection

Stoe Stadi 4 diffractometer  
 $\omega$ -2 $\theta$  scans  
 Absorption correction:  
 none  
 962 measured reflections  
 447 independent reflections  
 341 observed reflections  
 $[|F| > 3\sigma(|F|)]$   
*R<sub>int</sub>* = 0.081

$\theta_{\max}$  = 29.84°  
*h* = 0 → 14  
*k* = 0 → 5  
*l* = -9 → 9  
 2 standard reflections  
 frequency: 60 min  
 intensity variation:  
 2.2–2.7%

#### Refinement

Refinement on *F*  
*R* = 0.055  
*wR* = 0.043  
*S* = 1.75  
 341 reflections  
 24 parameters  
 $w = 1/\sigma^2(|F|)$   
 $(\Delta/\sigma)_{\max} = 0.0001$   
 $\Delta\rho_{\max} = 3.8 \text{ e } \text{Å}^{-3}$   
 $\Delta\rho_{\min} = -3.9 \text{ e } \text{Å}^{-3}$

Extinction correction:  
 Zachariasen (1968)  
 Extinction coefficient:  
 $g = 0.021 (3)$   
 Atomic scattering factors  
 from *International Tables*  
 for *X-ray Crystallography*  
 (1974, Vol. IV)

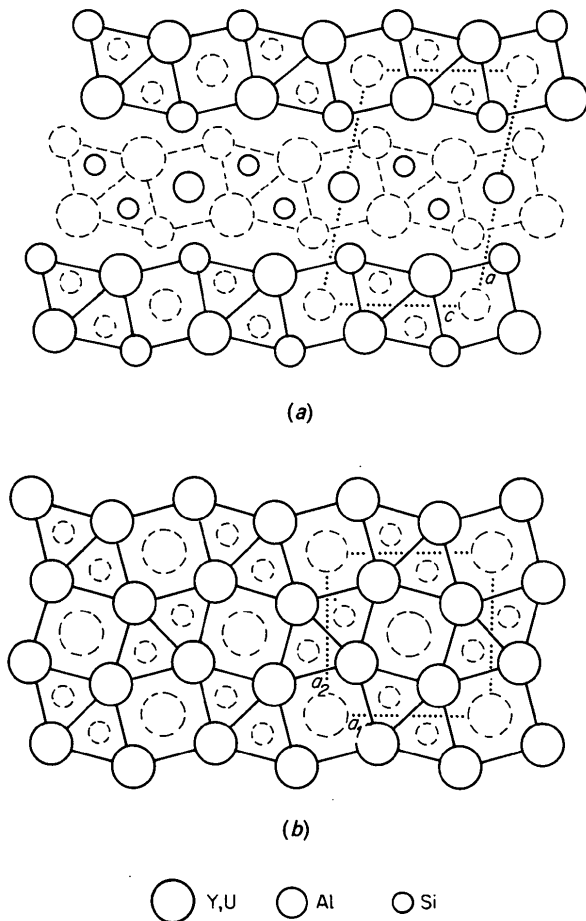


Fig. 2. (a) Structural projection of monoclinic Y<sub>2</sub>Al<sub>3</sub>Si<sub>2</sub> along [010] and (b) comparison with the projection of tetragonal U<sub>3</sub>Si<sub>2</sub> along [001]. Full lines and circles are at height ½, dashed lines and circles are at height 0.

Table 1. Fractional atomic coordinates and equivalent isotropic displacement parameters ( $\text{\AA}^2$ )
$$U_{\text{eq}} = (1/3) \sum_i \sum_j U_{ij} a_i^* a_j^* \mathbf{a}_i \cdot \mathbf{a}_j$$

		x	y	z	$U_{\text{eq}}$
Y	4 (i)	0.6202 (1)	0	0.3230 (2)	0.0082 (4)
Al(1)	4 (i)	0.3058 (4)	0	0.1342 (7)	0.014 (1)
Al(2)	2 (a)	0	0	0	0.016 (2)
Si	4 (i)	0.0931 (3)	0	0.6403 (6)	0.011 (1)

Table 2. Selected interatomic distances ( $\text{\AA}$ )

Y—Si	2.892 (4)	Al(1)—Y	3.209 (4)
Y—Si × 2	2.964 (3)	Al(1)—Y	3.260 (5)
Y—Si × 2	3.015 (3)	Al(1)—Y	3.521 (5)
Y—Al(2) × 2	3.019 (1)	Al(1)—Si	3.551 (5)
Y—Al(1) × 2	3.188 (4)	Al(2)—Si × 2	2.732 (4)
Y—Al(1)	3.209 (4)	Al(2)—Y × 4	3.019 (1)
Y—Al(1)	3.260 (5)	Al(2)—Al(1) × 2	3.076 (4)
Y—Al(1)	3.521 (5)	Al(2)—Al(1) × 4	3.081 (4)
Y—Y	3.713 (2)	Si—Si	2.382 (5)
Y—Y × 2	3.761 (2)	Si—Al(1) × 2	2.599 (3)
[Y—Y × 2	4.035 (1)]	Si—Al(2)	2.732 (4)
Al(1)—Si × 2	2.599 (3)	Si—Y	2.892 (4)
Al(1)—Al(1) × 2	2.780 (4)	Si—Y × 2	2.964 (3)
Al(1)—Al(2)	3.076 (4)	Si—Y × 2	3.015 (3)
Al(1)—Al(2) × 2	3.081 (4)	Si—Al(1)	3.551 (5)
Al(1)—Y × 2	3.188 (4)		

Structure solution was performed using *MULTAN87* (Debaerdemaeker, Germain, Main, Tate & Woolfson, 1987). Least-squares refinement included varying a scale factor, isotropic extinction, atomic positional and anisotropic displacement parameters. Programs used for data reduction and structure refinement were from the *Xtal3.2* system (Hall, Flack & Stewart, 1992).

Lists of structure factors, anisotropic displacement parameters and bond distances have been deposited with the IUCr (Reference: DU1080). Copies may be obtained through The Managing Editor, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

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*Acta Cryst.* (1994). **C50**, 1379–1381

## Synthetic Spheniscidite

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(Received 16 March 1994; accepted 11 April 1994)

## Abstract

Spheniscidite, ammonium hydroxide iron phosphate dihydrate,  $[\text{Fe}_2(\text{NH}_4)(\text{OH})(\text{PO}_4)_2] \cdot 2\text{H}_2\text{O}$ , is known as a mineral isotypic with leucophosphite. Crystals were obtained by hydrothermal synthesis at medium temperature. The crystal structure determination confirms the leucophosphite structure type and provides detailed localization of the water molecules and ammonium cations inside the cavities of the framework.

## Comment

With a view to synthesizing some oxyfluorinated phosphates of iron which have microporous frameworks, we investigated the system  $\text{FeO}(\text{OH})\text{—H}_3\text{PO}_4\text{—HF—1,3-diaminopropane}$  (hereafter 1,3-DAP)— $\text{H}_2\text{O}$  through the synthetic route described by Tapp, Milestone & Bibby (1988), which uses an organic amine as a template. Under our reaction conditions, 1,3-DAP decomposed and  $[\text{Fe}_2(\text{NH}_4)(\text{OH})(\text{PO}_4)_2] \cdot 2\text{H}_2\text{O}$ , which corresponds to the formula of spheniscidite (Wilson & Bain, 1986), was formed. The structure of mineral spheniscidite has not yet been described, but is claimed to be isotypic with leucophosphite  $[\text{KFe}_2(\text{PO}_4)_2(\text{OH})] \cdot 2\text{H}_2\text{O}$  (Moore, 1972). Related substances with this well known structure type, but different chemical compositions, can be found in the literature:  $\text{GaPO}_4 \cdot 2\text{H}_2\text{O}$  (Mooney-Slater, 1966), also labelled  $\text{GaPO}_4\text{—C}_7$  by Wang, Yang, Feng, Shang & Xu (1989), and  $\text{RbMo}_2\text{P}_2\text{O}_{10} \cdot 1\text{—}x\text{H}_2\text{O}$  (Leclaire, Borel, Grandin & Raveau, 1994). The detailed structure determination of spheniscidite is presented here.

The three-dimensional network of spheniscidite is built up from octahedrally coordinated  $\text{Fe}^{\text{III}}$  and  $\text{PO}_4$  tetrahedra. It can also be described as the corner sharing of octameric building units (Fig. 1a). This octamer consists of an edge-sharing bioctahedron of  $\text{Fe}^{\text{III}}$ , whose common edge is delimited by two OH groups which are also shared with the satellite octahedra. One vertex of each satellite octahedron is occupied by a terminal water molecule. The octamer is completed by four  $\text{PO}_4$  tetrahedra, each sharing either two or three corners with the Fe octahedra of the octamer. After their linkage, these octamers adopt a quasi-cc lattice within the unit